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## Negative magnetostrictive magnetoelectric coupling of BiFeO<sub>3</sub>

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How magnetoelectric coupling actually occurs on a microscopic level in multiferroic BiFeO<sub>3</sub> is not well known. By using high-resolution single crystal neutron diffraction techniques, we have determined the electric polarization of each individual element of BiFeO<sub>3</sub>, and concluded that magnetostrictive coupling suppresses the electric polarization at the Fe site below  $T_N$ . This negative magnetoelectric coupling appears to outweigh the spin current contributions arising from the cycloid spin structure, which should produce positive magnetoelectric coupling.

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Multiferroic materials with a coexistence of both ferroelectricity and magnetism offer a huge potential and, at the same time, pose new challenges for our understanding of how magnetism and ferroelectricity can be coupled to one another in real materials. Once fully achieved, this understanding of their coupling, the so-called magnetoelectric effects, can lead to better manipulation of unusual multiferroic behavior. However, one needs to carry out precise measurements of both the structure and dynamics in order to gain a fundamental understanding of the underlying physics. Despite the academic and technological importance, however, the origin of magnetoelectric coupling has proven to be often challenging to address experimentally for a given material.

BiFeO<sub>3</sub> is arguably one of the most extensively studied multiferroic materials with several distinctive properties.<sup>3–5</sup> For example, it has both magnetic and ferroelectric phase transitions above room temperature,  $T_N = 650 \text{ K}$  and  $T_C =$ 1100 K. Moreover, it has one of the largest reported values ever of polarization,  $\sim 86 \ \mu\text{C/cm}^2$ , with the noncentrosymmetric space group of R3c in the ferroelectric phase, as shown in Fig. 1(a). However, how magnetoelectric coupling actually occurs and moreover how to understand it remains to be resolved. Another interesting point to be noted is that when it undergoes basically a G-type magnetic ordering at 650 K, an incommensurate structure is formed with an extremely long period of 620 Å due to the Dzyaloshinski-Moriya (DM) interaction without breaking the crystal symmetry [see Figs. 1(b) and 1(c)].<sup>6</sup> In hexagonal notation, the propagation vector of the incommensurate structure is  $\mathbf{Q} = [0.0045, 0.0045, 0]$ at room temperature with a chiral vector of  $\mathbf{e}_3 = [-1 \ 1 \ 0]$ . The spin waves of BiFeO<sub>3</sub> measured by inelastic neutron scattering techniques<sup>7,8</sup> are consistent with a Heisenberg Hamiltonian with a DM interaction, which was derived from bulk measurements.9

As regards magnetoelectric coupling, multiferroic materials with chiral magnetic structures offer an interesting, as yet unexplored, possibility for inverse DM effects. This inverse DM effect induces an electric polarization when a particular chiral structure sets in. <sup>10</sup> The underlying mechanism of such additional polarization has been theoretically investigated by several groups. <sup>11,12</sup> Another equally viable scenario is the more classical mechanism of lattice-mediated spin-lattice coupling. Few systematic studies have so far been made in which one of the two mechanisms works for BiFeO<sub>3</sub>.

Here, we present detailed experimental measurements of magnetoelectric coupling in BiFeO<sub>3</sub> by using high-resolution single crystal neutron diffraction techniques. Through close examination of the temperature-dependent electric polarizations by Bi and Fe atoms, we demonstrate that magnetostrictive magnetoelectric coupling suppresses the polarization of Fe in the magnetically ordered phase, outweighing that due to the inverse DM effect.

High-resolution single crystal neutron diffraction experiments were carried out using a neutron wavelength of 0.835 Å from 300 to 850 K at the D9 diffractometer of the ILL, France. We used single crystals grown by a flux method with a typical size of  $1.6 \times 2.6 \times 2.4$  mm<sup>3</sup>. We have carried out all our analysis using the FULLPROF program. <sup>13</sup>

BiFeO<sub>3</sub> of the R3c space group undergoes an incommensurate antiferromagnetic transition at  $T_N = 650$  K with a cycloid magnetic structure as shown in Fig. 1(a). Because of the threefold rotational symmetry along the c axis, there are three equivalent propagation vectors for the cycloid magnetic structure with  $Q_1 = [1\ 1\ 0],\ Q_2 = [-2\ 1\ 0],$  and  $Q_3 = [1\ -2\ 0],$  and they are thermally populated as separate magnetic domains, as seen by polarized neutron diffraction experiments. There are two possible directions of spin rotation for each of the three Q vectors, which are related

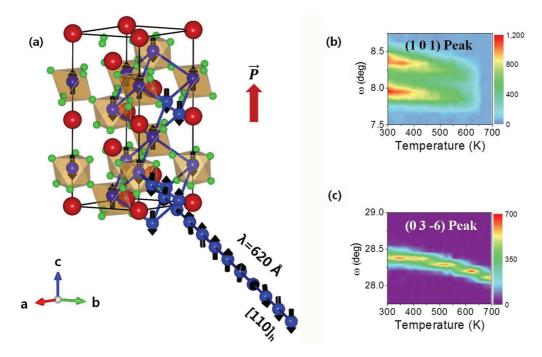


FIG. 1. (Color online) (a) Cycloid magnetic structure of BiFeO<sub>3</sub> with the propagation vector Q along the [1 1 0] direction. In this hexagonal setting, the total ferroelectric polarization (P) is pointing along the c axis. (b) and (c) show the temperature dependence of magnetic (1 0 1) and nuclear (0 3 -6) Bragg peaks.

to the aforementioned DM interaction. For example, the spin chiral vector  $e_1$  parallel to the  $[-1\ 1\ 0]$  axis produces a clockwise rotation of the cycloid structure while  $e_2$  parallel to the  $[1\ -1\ 0]$  axis does an anticlockwise rotation, when the cycloid structure is viewed on the (hhl) plane. A recent x-ray scattering experiment with polarization analysis shows that the sense of the spin rotation is clockwise.<sup>15</sup>

This clockwise spin rotation can then give rise to an induced electric polarization  $\Delta P$  via the inverse DM effect. This additional electric polarization can be written in the following form,  $\Delta P = Ae \times Q$ , where A is a material specific coefficient, e is the spin-rotation chiral vector, and Q is the propagation vector of the chiral magnetic structure. 12 Furthermore, it is important to note that a Ginzburg-Landau analysis based on the symmetry of BiFeO<sub>3</sub> predicts that  $\Delta P$ should be parallel to the total polarization direction in the paramagnetic phase, i.e., positive  $\Delta P$ . In our calculation, the total electric polarization (P) always points along the c axis above  $T_N$ . Notice that this dependence of  $\Delta P$  due to the inverse DM effect is equally valid for the other two Q vectors. On the other hand, magnetoelectric coupling of a magnetostrictive nature can produce negative  $\Delta P$ . Therefore, one can precisely determine which one of the two mechanisms is at work for BiFeO<sub>3</sub> simply by measuring  $\Delta P$  below  $T_N$ .

This realization opens up a simple yet elegant way of addressing the issue of magnetoelectric coupling in BiFeO<sub>3</sub>. In fact, we have tried to answer this question by carrying out high-resolution structure studies before, where we noted a quite significant change in the lattice constants below  $T_N$ . However, because of technical limitations in our previous diffraction experiments, we had to make most of our analysis then using synchrotron data. More specifically, we could not examine the temperature-dependent contributions to the

electric polarization by individual elements because of the lack of accurate information about the O positions over the entire temperature range: X-ray data are typically less sensitive to lighter elements such as O as compared to neutrons.

In order to overcome these previous technical difficulties, we have now carried out higher-resolution single crystal neutron diffraction experiments using the D9 beamline. We have taken special efforts to increase the number of nuclear peaks by a factor of 2 as compared with the previous experiment as well as to measure the data over a very wide temperature range, in particular, extending the temperature range above  $T_N$ .

Let us first explain the aim of the experiment from a crystallographic viewpoint. To estimate the individual electric polarization by Bi and Fe atoms, we have assumed the high temperature paraelectric phase of the Pm-3m space group. Although the ferroelectric transition and the space group of the paraelectric phase are still under debate, <sup>17,18</sup> one of the proposed candidates is the Pbnm orthorhombic structure (see Fig. 2), and our analysis and conclusion below are valid regardless of the nature of the ferroelectric transition since we are only concerned here with temperature-induced electric polarization. With the paraelectric phase of Pm-3m, the electric polarization by individual atoms can be calculated by simply measuring the relative shifts of Fe and Bi atoms with respect to the center of the oxygen octahedron. For more simplicity, we can take the relative shift of Fe and Bi atoms from their original positions in the paraelectric phase. Since the space group R3c has an uncertain origin from a crystallographic viewpoint, usually Bi(z) is fixed as an origin. However, in this Rapid Communication we consider the shift of Bi and Fe atoms relative from the oxygen atom, thus we fixed the z position of the oxygen as an origin (see Fig. 2).

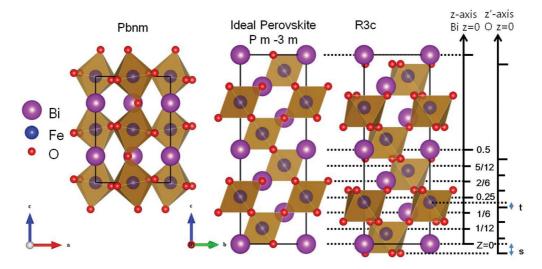


FIG. 2. (Color online) Left to right: Paraelectric *Pbnm* and ideal *Pm-3m* perovskite structure as well as ferroelectric *R3c* space group.

In the figure, "s" is the z component of the Bi atom, and then, equivalent to the shift of Bi from the cubic position, it is described as "sc." Meanwhile, "t" is the z component of the Fe atom measured from  $\frac{1}{4}$  position of the cubic phase and the shift of Fe is denoted by "tc." These two shifts of Fe (tc) and Bi (sc) can then be readily translated into the induced electric polarization of the respective elements by multiplying the nominal charges of Fe (3+) and Bi (3+). We note that, according to a theoretical calculation, 5 the Born effective charges of Fe and Bi are 3.49 and 4.37, respectively. Thus our discussion below can change quantitatively, but not qualitatively.

We have analyzed this observed data using a total of over 200 nuclear Bragg peaks. The representative data at two temperatures are given in Fig. 3 (see the Supplemental Material  $^{19}$  for the summary). We also plotted the temperature dependence of the lattice constants and the rotation angle ( $\omega$ ) of the oxygen octahedron along the c axis. The lines in Figs. 3(a) and 3(b) represent our calculated results using the Debye-Grüneisen formula with the same parameters as given in Ref. 16.

The relative shifts of Fe and Bi after being multiplied by the c lattice constant are shown in Fig. 3(d) as tc and sc. With these measured values of tc and sc, we estimate a total polarization value of  $\sim 73 \ \mu\text{C/cm}^2$  at room temperature. There are two items noteworthy in the figure. First, tc corresponding to the electric polarization of Fe is as large as half of sc corresponding to the electric polarization of Bi. This implies that both Fe and Bi almost equally contribute to the total polarization. Second, although the Bi shift increases monotonously upon cooling with no hint of an anomaly at  $T_N$ , there is a clear drop below  $T_N$  in the temperature dependence of the Fe shift (tc). This shift of the Fe position at  $T_N$  bears a certain similarity to the Mn off-centering observed in another multiferroic hexagonal manganite at its own antiferromagnetic transition.<sup>20</sup> The decreased electric polarization of Fe clearly then rules out the possibility of the magnetoelectric effect due to inverse DM effects, leaving a magnetostrictive origin as probably a most likely source for BiFeO<sub>3</sub>.

In order to further analyze the experimental data, we assume that each of the Fe  $(P_{\rm Fe})$  and Bi  $(P_{\rm Bi})$  electric polarizations follows the usual temperature dependence of first order in their respective Ginzburg-Landau (GL) free energies  $(F_{\rm Bi}$  and  $F_{\rm Fe})$  with a Curie temperature of 1100 K,  $F_{\rm Bi/Fe} = \frac{\alpha_2}{2} P_{\rm Bi/Fe}^2 + \frac{\alpha_4}{4} P_{\rm Bi/Fe}^4 + \frac{\alpha_6}{6} P_{\rm Bi/Fe}^6$ . In addition, we add a magnetoelectric coupling term of magnetostrictive origin  $(F_{\rm ME})$  to the total GL free energy,  $F_{\rm ME} = \delta M^2 P_{\rm Fe}^2$ . We took the measured temperature-dependent intensity of the  $(0\ 0\ 3) \pm Q$  magnetic superlattice peak as our M in the GL functional: The magnetic superlattice peaks could not be separated because of the instrumental resolution. By using this full GL free energy  $(F_{\rm total} = F_{\rm Bi} + F_{\rm Fe} + F_{\rm ME})$  we have tried to fit the temperature dependence of the relative shifts (tc and sc) of Fe and Bi.

First of all, using the GL free energy functional with only  $F_{\rm Bi}$  and  $F_{\rm Fe}$  terms, we have fitted the temperature dependence of both Fe and Bi shifts (tc and sc) above  $T_N$ , as shown in Fig. 3(d). In the case of the Bi shift (sc), the experimental data follow well this theoretical line across and even below  $T_N$ , indicating that the Bi electric polarization  $(P_{Bi})$  is not affected by the emergence of antiferromagnetic order. However, the Fe shift (tc) shows a clear deviation from the theoretical line (dashed line) estimated from above  $T_N$ : This change in the Fe shift below  $T_N$  is denoted as  $\Delta tc$  in Fig. 3(d). This downwards deviation of tc and so a reduction in the Fe electric polarization  $(P_{\rm Fe})$  observed below  $T_N$  imply two items of upmost importance. First and foremost, there is an unequivocal coupling between the Fe electric polarization  $(P_{\text{Fe}})$  and the magnetic moment (M). Second, the Fe electric polarization is suppressed by the onset of the magnetic transition with respect to the values expected of an otherwise nonmagnetic and ferroelectric phase. In order to demonstrate these points, we carried out a further analysis now using the full GL free energy functional  $(F_{\text{total}})$ . As shown by the solid line for the Fe shift (tc) below  $T_N$ , this approach is clearly successful in explaining the anomaly seen in the temperature dependence of the Fe shift (tc).

As further proof of the existence of the coupling between  $P_{\rm Fe}$  and M, we now plot the additional electric polarization of Fe ( $\Delta P_{\rm Fe}=q\,\Delta {\rm tc}$ , where q=+3e for Fe<sup>3+</sup>) against the

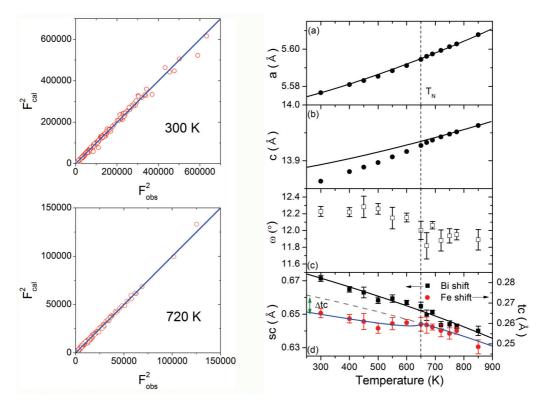


FIG. 3. (Color online) Left: Calculated and observed squared structure factor at two representative temperatures of 300 and 720 K with 222 and 218 nuclear Bragg peaks, respectively. Right: The temperature dependence is shown of the lattice constants [(a) and (c)] together with the antiphase rotation angle  $\omega$  of the oxygen octahedron in addition to the Bi (sc) and Fe (tc) shifts along the c axis. The solid lines in (a) and (b) represent our theoretical calculations using a Debye-Grüneisen formula as in the text, while the solid lines in (d) represent the theoretical temperature dependence of the first-order ferroelectric order parameter based on the Ginzburg-Landau free energy analysis with the magnetoelectric coupling term, as discussed in the text. The dashed line plots the theoretical calculation results of the Fe shift (tc) without the magnetoelectric term.

measured intensity of the  $(0\ 0\ 3) \pm Q$  magnetic superlattice peak in Fig. 4. With this information, we also reanalyzed our previous data taken from another single crystal diffractometer (FONDER), and present them in Fig. 4 together with the theoretical line as discussed above. As can be seen, both sets of our

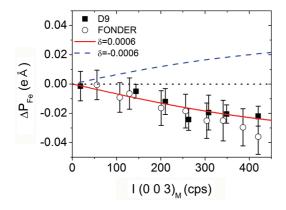


FIG. 4. (Color online) Plot of the induced electric polarization of Fe ( $\Delta P_{\rm Fe}$ ) against the measured intensity of the (0 0 3)  $\pm$  Q magnetic superlattice peaks. The solid line represents the theoretical calculation results based on the Ginzburg-Landau free energy analysis with a negative magnetoelectric coupling as discussed in the text, while the dashed line shows the theoretical results expected for the case with the opposite sign for the magnetoelectric coupling.

data from the two different instruments display  $\Delta P_{\rm Fe}$ , which is in good agreement with one another. It is also consistent with the theoretical calculations of the GL free energy with negative magnetostrictive magnetoelectric coupling, i.e., a positive sign of  $\delta$  in our Ginzburg-Landau functional. We should note that our results cannot be reconciled by theoretical calculations with an opposite sign of magnetoelectric coupling, as shown by the dashed line in Fig. 4.

We note that our observation of the overall negative magnetoelectric coupling is qualitatively consistent with the field-induced electric polarization observed experimentally.  $^{16,21,22}$  Notice that all previous pyroelectric current measurements only measured the absolute value of  $\Delta P$ , not its sign for the intrinsic technical problems. However, we acknowledge there is a discrepancy at a quantitative level. For example, our estimated value of  $\Delta P$  is  $\sim\!400$  nC/cm² while the bulk value was reported to be  $\sim\!40$  nC/cm²  $^{16,21}$  It may well be plausible that our experimental values of the electric polarization might overestimate the magnetoelectric effect. We make one passing comment which has much wider implications: Our works and the conclusion made here suggest that it should be a very useful exercise for one to examine the origin of magnetoelectric coupling by using high-resolution diffraction studies as we have done for BiFeO3 here.

In summary, we have experimentally determined the origin of magnetoelectric coupling in BiFeO<sub>3</sub> by high-resolution

single crystal neutron diffraction studies. Our calculation of the induced electric polarization of each Fe and Bi in the magnetic phase demonstrates that magnetoelectric coupling of the magnetostrictive origin suppresses the electric polarization at the Fe site below  $T_N$ , outweighing that of the inverse Dzyaloshinskii-Moriya effects, and becomes the dominant magnetoelectric coupling mechanism for BiFeO<sub>3</sub>.

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